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### ㉒ Epitaxial growth process and growing apparatus.

㉓ A process of epitaxially growing a semiconductor Si, Ge or SiGe single crystal layer on a semiconductor (Si or Ge) single crystal substrate, comprising the steps of: allowing a raw material gas (e.g., Si<sub>2</sub>H<sub>6</sub>, GeH<sub>4</sub>) for the layer and a fluoride gas (e.g., Si<sub>2</sub>F<sub>6</sub>, GeF<sub>4</sub>, BF) of at least one element selected from the group consisting of the semiconductor element of the layer and a dopant for the layer to simultaneously flow over the substrate; and applying an ultraviolet light to the substrate to decompose the gases by an ultraviolet light excitation reaction to deposit the layer on the surface of the substrate heated at a temperature of from 250 to 400 °C.

Prior to the epitaxial growth of the semiconductor layer, the substrate is cleaned by allowing the fluoride gas to flow over the substrate having a temperature of from a room temperature to 500 °C, and by irradiating an ultraviolet light to the substrate to remove a natural oxide layer from the substrate surface.

EP 0 368 651 A2

## EPITAXIAL GROWTH PROCESS AND GROWING APPARATUS

## BACKGROUND OF THE INVENTION

## 1. Field of the Invention

The present invention relates to an epitaxial growing of a semiconductor single crystal layer, and more particularly, to a process of forming an epitaxially grown layer on a single crystal silicon substrate by photo-decomposing a gas raw material with an irradiation of ultraviolet laser light or lamp light, and an apparatus for carrying out the process. The present invention is applied to an epitaxial growth of a silicon (Si), germanium (Ge) or silicon-germanium (SiGe) single crystal layer (thin film).

## 2. Description of the Related Art

In a production of semiconductor devices such as bipolar transistors and MOS field effect transistors (FETs), an epitaxial growth temperature must be lowered when growing a single crystal epitaxial layer of Si, Ge or SiGe, to avoid the generation of lattice defects, and variations in the impurity diffusion profile in a single crystal (e.g., Si) substrate (wafer), to enable the miniaturization of a semiconductor device, such as a very-large-scale integration (VLSI) device.

Recently, instead of a thermal decomposition (hydrogen reduction) process having a growing temperature of 950°C or more, a photo CVD process using a photoexciting reaction has been proposed for an epitaxial growth at a temperature lower than the above-mentioned CVD processes. In this case, the epitaxial growth of, e.g., Si, is performed at a temperature of from 600 to 900°C. The photo CVD process however, requires a high temperature annealing treatment (at 900°C or more) in an ultrahigh vacuum or a hydrogen atmosphere, for removing (cleaning) a natural oxide layer formed on the substrate surface prior to the epitaxial growth, and requires an addition of hydrogen gas to prevent the formation of an oxide layer during the epitaxial growth.

When the high temperature annealing treatment for removing the natural oxide layer is applied to a substrate in which impurity doped (diffused) regions have been formed, the regions i.e., diffusion profiles) are undesirably expanded. Since the use of hydrogen gas is potentially dangerous (e.g., an explosion could occur) in the operation of an epitaxial growth apparatus, it is preferable to avoid

the use of hydrogen gas.

To avoid the high temperature annealing treatment, it has been proposed that SiH<sub>2</sub>F<sub>2</sub> gas be added in the hydrogen gas to generate SiF radicals therefrom at a low temperature of 100 to 300°C, as these radicals accelerate the removal of the natural oxide layer. Furthermore, the addition of SiH<sub>2</sub>F<sub>2</sub> gas to the gas raw material generates SiF radicals which prevent the formation of the oxide layer during the epitaxial growth (cf. e.g., A. Yamada et al. Extended Abstracts of the 18th Conference on Solid State Devices and Materials, Tokyo, 1986, pp.217 - 220). SiH<sub>2</sub>F<sub>2</sub> gas, however, has a spontaneous combustion property and is potentially more dangerous than hydrogen, and therefore, it is necessary to avoid the use of SiH<sub>2</sub>F<sub>2</sub> gas.

## SUMMARY OF THE INVENTION

An object of the present invention is to further reduce the epitaxial growth temperature in the photo CVD process, and the cleaning temperature for the substrate surface, by using a safe gas instead of the hydrogen gas and SiH<sub>2</sub>F<sub>2</sub> gas.

Another object of the present invention is to provide an improved process of epitaxially growing a semiconductor (Si, Ge or SiGe) single crystal layer (thin film) by a photo CVD process at a temperature lower than used in a conventional photo CVD process.

Still another object of the present invention is to provide an apparatus for carrying out the improved epitaxial growth process.

The above-mentioned and other objects of the present invention are attained by providing a process of epitaxially growing a semiconductor single crystal layer on a semiconductor single crystal substrate, comprising the steps of: allowing a raw material (hydride) gas for said layer and a fluoride gas of at least one element selected from the group consisting of the semiconductor element of said layer and a dopant for said layer to simultaneously flow over said substrate; and applying an ultraviolet light to said substrate to decompose said gases by an ultraviolet light excitation reaction to deposit said layer on the surface of said substrate heated at a temperature of from 250 to 400°C.

The semiconductor single crystal layer to be epitaxially grown is of Si, Ge or SiGe and is deposited on a semiconductor single crystal substrate of Si or Ge.

Preferably the raw material hydride gas for the grown layer is of silicon hydride [Si<sub>n</sub>H<sub>2n+2</sub> (n = 2, 3, 4)] for the Si layer or is of germanium hydride

[ $\text{Ge}_n\text{H}_{2n+2}$  ( $n = 1, 2, 3$ )] for the Ge layer. It is possible to use a gas mixture of the silicon hydride gas and the germanium hydride gas as the raw material hydride gas for the SiGe layer.

It is preferable to use at least one fluoride gas of silicon fluoride (and/or germanium fluoride), boron fluoride, arsenic fluoride, and phosphorus fluoride. Note, where the Si epitaxial layer is grown, the germanium fluoride gas should not be used, and where the Ge epitaxial layer is grown, the silicon fluoride gas should not be used.

When the ultraviolet light irradiates the silicon hydride gas (and/or germanium hydride gas), the gas is decomposed to deposit Si (and/or Ge) on the Si or Ge substrate, with the result that the Si (and/or Ge) layer is epitaxially grown. When the ultraviolet light irradiates the fluoride gas of Si, Ge, B, As and/or P, the gas is decomposed to generate fluoride radicals of  $\text{SiF}$ ,  $\text{GeF}$ ,  $\text{BF}$ ,  $\text{AsF}$  and/or  $\text{PF}$  which remove oxide to the inclusion of the oxide in the growing layer during the epitaxial growth period. This oxide is usually generated by a reaction with oxygen remaining within a reactor and is diffused through walls of the reactor and other parts of the apparatus. Furthermore, the fluoride radicals can be decomposed to generate elements of Si, Ge, B, As and P. The elements of Si and Ge contribute to the growth of the epitaxial layer, and the elements of B, As and P are contained in the epitaxial layer and serve as dopants for determining the conductivity type.

The addition of the fluoride gas in the raw material gas in the photo CVD process for epitaxial growth enables the epitaxial temperature to be made lower than that used in a conventional photo CVD process.

Prior to the above-mentioned epitaxial growth of the Si, Ge or SiGe layer, the surface of the semiconductor single crystal substrate should be cleaned, i.e., a natural oxide layer formed on the substrate surface should be removed (etched out) by allowing the above-mentioned fluoride gas to flow over the substrate kept at a temperature of from a room temperature to 500°C and by irradiating the ultraviolet light onto the fluoride gas to decompose the gas by photoexcitation. Namely, the decomposition of the fluoride gas generates the fluoride radicals which remove (etch) the natural oxide layer. The silicon fluoride ( $\text{SiF}_4$ ,  $\text{Si}_2\text{F}_6$ ) gas, germanium fluoride ( $\text{GeF}_4$ ) gas, boron fluoride ( $\text{BF}_3$ ) gas, arsenic fluoride ( $\text{AsF}_3$ ,  $\text{AsF}_5$ ) gas, and phosphorus fluoride ( $\text{PF}_3$ ,  $\text{PF}_5$ ) gas used instead of the  $\text{SiH}_2\text{F}_2$  gas are incombustible, and thus are safer than the  $\text{SiH}_2\text{F}_2$  gas. The germanium fluoride gas should not be used, prior to the epitaxial growth of the Si layer, and the silicon fluoride gas should not be used prior to the epitaxial growth of the Ge layer.

Preferably, the ultraviolet light has a wavelength of from 250 to 150 nm, suitable for photoexciting the above-mentioned gases, and is irradiated by a laser or a high pressure mercury lamp.

According to the present invention, the fluoride radicals such as  $\text{SiF}$  radicals are utilized in the epitaxial growth and the cleaning of the substrate surface. If the radical generation is interrupted, between the cleaning step and the epitaxial growth step, a natural oxide layer may be formed on the substrate to prevent the epitaxial growth. Therefore, in practice, such an interruption must be avoided by continuously feeding the fluoride gas and by adding the raw material hydride gas to the flowing fluoride gas during the epitaxial growth period. Since, however, the gas flow rate is usually controlled by a mass flow controller, several seconds must pass before reaching a preset value of the flow rate, from the start of the flow of the raw material hydride gas. Taking this time-lag into consideration, the raw material hydride gas is fed at a flow rate larger than the preset value for several seconds from the start of the flow, and as a result, the deposition rate of the semiconductor single crystal layer may be temporarily increased; in the worst case, the epitaxial growth may be hampered.

Another object of the present invention is to provide an epitaxial growth apparatus by which a procedure from the substrate surface cleaning to the photo CVD epitaxial growth is stably carried out without difficulty.

The above-mentioned object is attained by providing an apparatus for epitaxially growing a semiconductor single crystal layer on a semiconductor single crystal substrate, comprising a reactor having a susceptor for said substrate and a window allowing a passage of ultraviolet light; an ultraviolet light source; a first inlet pipe for feeding a fluoride gas, which pipe is provided with a spray end portion thereof located above and close to the substrate; a second inlet pipe for feeding a raw material gas for the layer, which pipe is branched into a first branch pipe having a first valve and joined to said first inlet pipe, and a second branch pipe having a second valve and introduced into a region within the reactor not irradiated with the ultraviolet light.

In the apparatus, the spray end portion for feeding the gases used for the cleaning and the epitaxial growth is placed above and close to the substrate, to generate a slight pressure difference between that in the vicinity of the substrate and other regions within the reactor chamber.

During the cleaning and epitaxial growth processes, the raw material gas is first fed into the not irradiated region to adjust and stabilize the flow rate thereof, and is then fed to the close vicinity of the substrate together with the fluoride gas, by

closing the second valve and simultaneously opening the first valve, whereby a stable photo CVD epitaxial growth is realized.

#### BRIEF DESCRIPTION OF THE DRAWINGS

The present invention will become more apparent from the description of the preferred embodiments set forth below with reference to the accompanying drawings, in which:

Fig. 1 is a schematic view of an epitaxial growth apparatus in a photo CVD system according to the present invention;

Fig. 2 is a schematic sectional view of a hetero-junction bipolar transistor; and

Fig. 3 is a schematic sectional view of another bipolar transistor

#### DESCRIPTION OF THE PREFERRED EMBODIMENTS

Referring to Fig. 1, an epitaxial growth apparatus having an improved pipeline system is used for epitaxially growing (depositing, forming) a semiconductor single crystalline layer on an semiconductor single crystal substrate by utilizing a photoexcitation reaction caused by an ultraviolet irradiation in accordance with the present invention.

The apparatus comprises a vacuum reactor chamber 2 with a transparent window 1 allowing a passage of ultraviolet light 12, and an ultraviolet light generator 3 (e.g., ArF excimer laser, a light pressure mercury lamp or the like). The reactor chamber 2 is communicated to a vacuum exhaust system (e.g., vacuum pump) 4 and is provided with an X-Y stage 5, a heating susceptor 6, and a pipe guide 7 for the ultraviolet light within the chamber 2. Gas inlet pipes 8 and 9 for feeding an inert gas (e.g., nitrogen (N<sub>2</sub>) gas) are fixed to the reactor chamber 2. A first inlet pipe 22 for feeding a fluoride gas (e.g., Si<sub>2</sub>F<sub>6</sub>, GeF, BF or the like) and having a mass flow controller 23 is set such that a spray end portion 24 thereof is located above and close to a semiconductor substrate (wafer) 11 set on the susceptor 6. The spray end portion 24 is provided with spraying pores distributed in a space corresponding to the size of the substrate 11 and arranged at a position at which it will not obstruct the ultraviolet irradiation. A second inlet pipe 15 for feeding a material hydride gas (e.g., Si<sub>2</sub>H<sub>6</sub>, GeH) is provided with a mass flow controller 16 and is branched, downstream of the controller 16, in a first branch pipe 17 with a first valve 19 and a second branch pipe 18 with a second valve 20. The first branch pipe 17 is joined to the first inlet pipe 22, to spray the hydride gas together with the fluoride

gas through the spray end portion 24. The end portion of the second branch pipe 18 is open in a region within the reactor chamber 2 not irradiated with the ultraviolet light, to provide a bypass feed of the hydride gas into the reactor chamber 2.

#### Example 1

According to the process of the present invention, a Si single crystalline layer is formed on a Si single crystalline substrate 11 in the following manner.

First, the Si substrate 11 is placed on the heating susceptor 6, the reactor chamber 2 is then exhausted by the vacuum exhaust system 4 to generate a vacuum and pressure of approximately  $1 \times 10^{-6}$  Torr, and nitrogen (N<sub>2</sub>) gas having a flow rate of 10 SCCM is continuously introduced into the reaction chamber 2 through the pipes 8 and 9 to form an inert gas atmosphere having a pressure of 4 Torr.

The Si substrate 11 is heated and kept at a temperature of 400 °C and silicon fluoride (Si<sub>2</sub>F<sub>6</sub>) gas is continuously fed at a flow rate of 15 SCCM through the first inlet pipe 22 and the spray end portion 24. Under the above-mentioned conditions, an ArF excimer laser device 4 as the ultraviolet light generator irradiates a laser ray (wavelength: 193 nm) 12 over the whole surface of the Si substrate 11 through a transparent window 1. The laser irradiation is performed for 10 minutes to etch (remove) a natural oxide layer formed on the Si substrate surface by a reduction of SiF radicals generated from the Si<sub>2</sub>F<sub>6</sub> gas, whereby the surface of the Si substrate 11 is cleaned.

Just before the end of the surface cleaning process, Si<sub>2</sub>H<sub>6</sub> gas as a raw material gas is passed into a not irradiated region within the reactor chamber 2 through the second branch pipe 18 from the second inlet pipe 15, by opening the second valve 20, at a flow rate of 0.2 SCCM. At this time, since the Si<sub>2</sub>F<sub>6</sub> gas flows out of the spray end portion 24, the introduced Si<sub>2</sub>H<sub>6</sub> gas does not flow on the Si substrate 11. When the flow rate of the Si<sub>2</sub>H<sub>6</sub> gas becomes stable, the second valve 20 is closed, and simultaneously the first valve 19 is opened, to feed the Si<sub>2</sub>H<sub>6</sub> gas into the first inlet pipe 22 to flow together with the Si<sub>2</sub>F<sub>6</sub> gas onto the Si substrate 11. When the Si<sub>2</sub>H<sub>6</sub> gas is irradiated with the ArF excimer laser ray 12 and decomposed by photo excitation, Si is deposited on the cleaned surface of the Si substrate to epitaxially grow a Si single crystal layer thereon. This epitaxial growth of Si after the cleaning forms an Si layer having a thickness of approximately 0.4 μm, in 10 minutes. The single crystallizability of the formed Si layer is confirmed by an electron channeling pattern (ECP)

technique. During the epitaxial growth, it is possible to prevent an inclusion of oxide contaminants (particles) generated by a reaction of Si with oxygen diffused out of the walls of the reactor chamber 2 in the growing Si layer, by both the reduction effect of SiF radicals and the direct reduction effect of ultraviolet irradiation on the oxide particles. The SiF radicals are generated not only directly by a photo-decomposition of the ultraviolet light of Si<sub>2</sub>F<sub>6</sub> gas but also indirectly by the effect of other radicals generated by the photo-decomposition of Si<sub>2</sub>H<sub>6</sub> gas.

#### Example 2

In the formation process of the Si layer of the above-mentioned Example 1, a Ge single crystal substrate is used instead of the Si single crystal substrate. Namely, the cleaning process and the epitaxial growth process are performed under the same conditions as Example 1, but the semiconductor material of the substrate is different. As a result, an Si single crystal layer is obtained (epitaxially grown).

#### Example 3

In the formation process of the Si layer of the above-mentioned Example 1, a mixture of Si<sub>2</sub>H<sub>6</sub> gas (50 vol%) and Si<sub>3</sub>H<sub>8</sub> gas (50 vol%) is used instead of Si<sub>2</sub>H<sub>6</sub> gas only, as the raw material gas. In this case, the Si single crystal layer is also obtained (epitaxially grown) on the Si substrate.

#### Example 4

In the formation process of the Si layer of Example 1, a boron fluoride (BF<sub>2</sub>) gas is used instead of the Si<sub>2</sub>F<sub>6</sub> gas. In this case, the Si single crystal layer is obtained (epitaxially grown) on the Si substrate and contains a boron dopant therein, and thus has a p-type conductivity.

#### Comparative Example A

An Si layer is formed on a Si single crystal substrate under the same conditions as those of the above-mentioned Example 1 except that the cleaning process is omitted. In this case, the Si substrate is heated at 400 °C and the Si<sub>2</sub>F<sub>6</sub> gas (15 SCCM) and Si<sub>2</sub>H<sub>6</sub> gas (0.2 SCCM) are simultaneously fed through the spray end portion 24. Thereafter, the ArF excimer laser ray 12 is irradiated to deposit Si and to form a Si layer on the Si

substrate 11. When the formed Si layer was examined by the EPC technique, it was confirmed that the Si layer was not a single crystal layer.

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#### Example 5

According to the process of the present invention, a Ge single crystal layer is formed on a Si single crystal substrate by using the apparatus shown in Fig. 1, in the following manner.

First, the Si substrate 11 is placed on the heating susceptor 6, then reactor chamber 2 is exhausted to reach a pressure of approximately 1 x 10<sup>-6</sup> Torr, and N<sub>2</sub> gas is continuously fed at a flow rate of 10 SCCM into the reaction chamber 2 to form an inert gas atmosphere at a pressure of 4 Torr. The Si substrate 11 is heated and held at a temperature of 250 - 400 °C, and germanium fluoride (GeF<sub>4</sub>) gas is continuously fed at a flow rate of 5 SCCM through the spray end portion 24. The ArF laser ray 12 is incident on the Si substrate 11 through the window 1 to remove a natural oxide layer from the Si substrate surface, for 60 minutes.

Next, GeH<sub>4</sub> gas (0.1 SCCM) is first fed into a not irradiated region in the reaction chamber 2 through the second branch pipe 18, and then fed into the first inlet pipe 2 to flow GeH<sub>4</sub> gas and GeF<sub>4</sub> gas over the Si substrate 11 through the spray end portion 24 by controlling the first and second valves 19 and 20, as explained in Example 1. As a result, GeH<sub>4</sub> gas is photo-excited to be compressed, and thus Ge is deposited on the cleaned Si substrate surface to epitaxially grow a Ge single crystal layer having a thickness of approximately 10 nm, in 10 minutes. The single crystallinity of the formed Ge layer is conformed by the ECP technique.

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#### Example 6

In the formation process of Ge layer of the above-mentioned Example 5, a Ge single crystalline substrate is used instead of the Si substrate. In this case, the Ge single crystal layer is also obtained (epitaxially grown).

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#### Comparative Example B

A Ge layer is formed on the Si substrate under the same conditions as those of Example 4, except that the cleaning process is omitted. In this case, the Si substrate is heated at 250 - 400 °C, and the GeF<sub>4</sub> gas (5 SCCM) and GeH<sub>4</sub> gas (0.1 SCCM) are simultaneously fed through the spray end portion 24. Thereafter, the ArF excimer laser ray 12 is

irradiated to deposit Ge and form the Ge layer on the Si substrate 11. When the formed Ge layer was examined by the EPC technique, it was confirmed that the Ge layer was not a single crystalline layer.

#### Example 7

According to the process of the present invention, an SiGe single crystal layer is formed on an Si single crystal substrate by using the above-mentioned epitaxial growth apparatus, in the following manner.

The Si substrate 11 is placed on the heating susceptor 6, the reactor chamber 2 is then exhausted, and N<sub>2</sub> gas is continuously fed at a flow rate of 10 SCCM into the reactor chamber 2 to form an inert atmosphere at a pressure of 4 Torr. The Si substrate 11 is heated and held at 250 - 400 °C and a mixed fluoride gas (5 SCCM) of Si<sub>2</sub>F<sub>6</sub> gas (50 vol%) and GeF<sub>4</sub> gas (50 vol%) is continuously fed through the spray end portion 24. A high pressure mercury lamp ("suitable?") 3 irradiates an ultraviolet light 12 on the Si substrate 11 through the window 1 to remove a natural oxide layer from the Si substrate surface, for 60 minutes.

Next, a mixed hydride gas (5 SCCM) of Si<sub>2</sub>H<sub>6</sub> gas (50 vol%) and GeH<sub>4</sub> gas (50 vol%) is fed into a not irradiated region through the second branch pipe 18 and then fed into the first inlet pipe 22 to pass the mixed gas of the above-mentioned four gases over the Si substrate 11 through the spray end portion 24 by controlling the valves 19 and 20, as explained in Example 1. As a result, the hydride gases are photoexcited to be decomposed, whereby Si and Ge are deposited on the clean Si substrate surface to epitaxially grow the SiGe single crystal layer having a thickness of approximately 10 nm, in 10 minutes. The formed SiGe layer is confirmed by the EPC technique to be a single crystal.

#### Comparative Example C

A SiGe layer is formed on the Si substrate under the same conditions as those of Example 7 except that the cleaning process is omitted. The formed SiGe layer is examined by the EPC technique to confirm that the SiGe layer is not a single crystal layer.

As mentioned above, the cleaning process of the substrate surface and the epitaxial growth process of the Si, Ge or SiGe layer are performed at a temperature lower than in a conventional case, by using a specific fluoride gas and hydride gas without hydrogen gas and utilizing ultraviolet irradiation to cause a photoexcitation of the gases.

The semiconductor single crystal layer epitaxially grown in accordance with the process of the present invention can be used in semiconductor devices, e.g., bipolar transistors, as shown in Figs. 2 and 3.

In Fig. 2, a hetero-junction bipolar transistor (HBT) comprises an Si single crystal collector substrate 26, an SiGe single crystal base layer 27, and a Si single crystal emitter layer 28. The SiGe layer 27 is epitaxially grown on the Si substrate 26 in a similar manner to that of Example 7, according to the present invention. The Si layer 28 is also epitaxially grown on the SiGe layer 27, by stopping the feeding of the germanium fluoride gas and germanium hydride gas and by continuously feeding the silicon fluoride gas and silicon hydride gas under the ultraviolet light irradiation. After the formation of the Si layer 28, the Si layer 28 and the SiGe layer 27 are selectively etched to form a mesa portion of the Si layer 28 and a portion of the SiGe layer 27, as shown in Fig. 2. Then suitable insulating layers and electrodes are formed in a conventional manner.

In Fig. 3, a bipolar transistor comprises an n-type Si single crystal collector substrate 31, a p-type Si single crystal base layer 32A, and an n-type impurity doped emitter region 33. The Si substrate 31 is selectively and thermally oxidized to form a field oxide (SiO<sub>2</sub>) layer 34. Then the Si layer 32A is epitaxially grown on the Si substrate 31 in a similar manner to that of Example 1 according to the present invention. In this case, a boron fluoride (BF<sub>2</sub>) gas is added to the Si<sub>2</sub>F<sub>6</sub> gas for p-type doping, and at the same time, a Si polycrystal layer 32B is formed on the oxide layer 34. The Si polycrystalline layer 32B is patterned by a conventional selective etching process to form a conductive line for a base electrode. An insulating layer 35 is formed over the whole surface and selectively etched to open an emitter contact hole, and As (or P) ions are doped in the Si layer 32A through the emitter contact hole by an ion-implantation process to form the n-type emitter region 33. The insulating layer 35 is selectively etched to open a base contact hole. Then a metal (Al-Si-Cu alloy) is formed over the whole surface and is patterned to form an emitter electrode 36, a base electrode 37, and a collector electrode indicated by a symbol "C".

It will be obvious that the present invention is not restricted to the above-mentioned embodiments and that many variations are possible for persons skilled in the art without departing from the scope of the invention.

#### Claims

1. A process of epitaxially growing a semiconductor single crystal layer on a semiconductor single crystal substrate, comprising the steps of: allowing a gas comprising the raw material for the layer and a fluoride gas to flow simultaneously over the substrate; and decomposing the gases by ultraviolet light excitation, to deposit the layer on the surface of the substrate.

2. A process according to claim 1, wherein the substrate is heated at 250 to 400 °C.

3. A process according to claim 1 or claim 2, wherein the substrate is silicon or germanium.

4. A process according to any preceding claim, wherein the layer is silicon, germanium or silicon-germanium.

5. A process according to claim 4, wherein the raw material gas is a silicon hydride of the formula  $Si_nH_{2n+2}$  (n = 2, 3, 4) or a germanium hydride of the formula  $Ge_mH_{2m+2}$  (m = 1, 2, 3), or a mixture thereof.

6. A process according to claim 4 or claim 5, wherein the fluoride gas is silicon fluoride (if the layer comprises silicon), germanium fluoride (if the layer comprises germanium), boron fluoride, arsenic fluoride and phosphorus fluoride.

7. A process according to any preceding claim, comprising the further steps, prior to the deposition of the layer, of cleaning the substrate surface by allowing the fluoride gas to flow over the substrate, and applying ultraviolet light to decompose the fluoride gas.

8. A process according to claim 7, wherein the substrate is maintained at ambient temperature to 500 °C.

9. A process according to any preceding claim, wherein the ultraviolet light has a wavelength of from 250 to 150 nm.

10. A process according to claim 9, wherein the ultraviolet light is irradiated from a laser or from a high-pressure mercury lamp.

11. Apparatus for epitaxially growing a semiconductor single crystal layer on a semiconductor single crystal substrate, comprising:  
a reactor having a susceptor for said substrate and a window allowing passage of ultraviolet light;  
an ultraviolet light source;  
a first inlet pipe for feeding a fluoride gas, which pipe is provided with a spray end portion thereof located above and close to the substrate; and  
a second inlet pipe for feeding a gas comprising the raw material for the layer, which branches into a first branch pipe having a first valve and joined to the first inlet pipe, and a second branch pipe having a second valve and leading into a region within the reactor not irradiatable with the ultraviolet light.

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Fig. 1

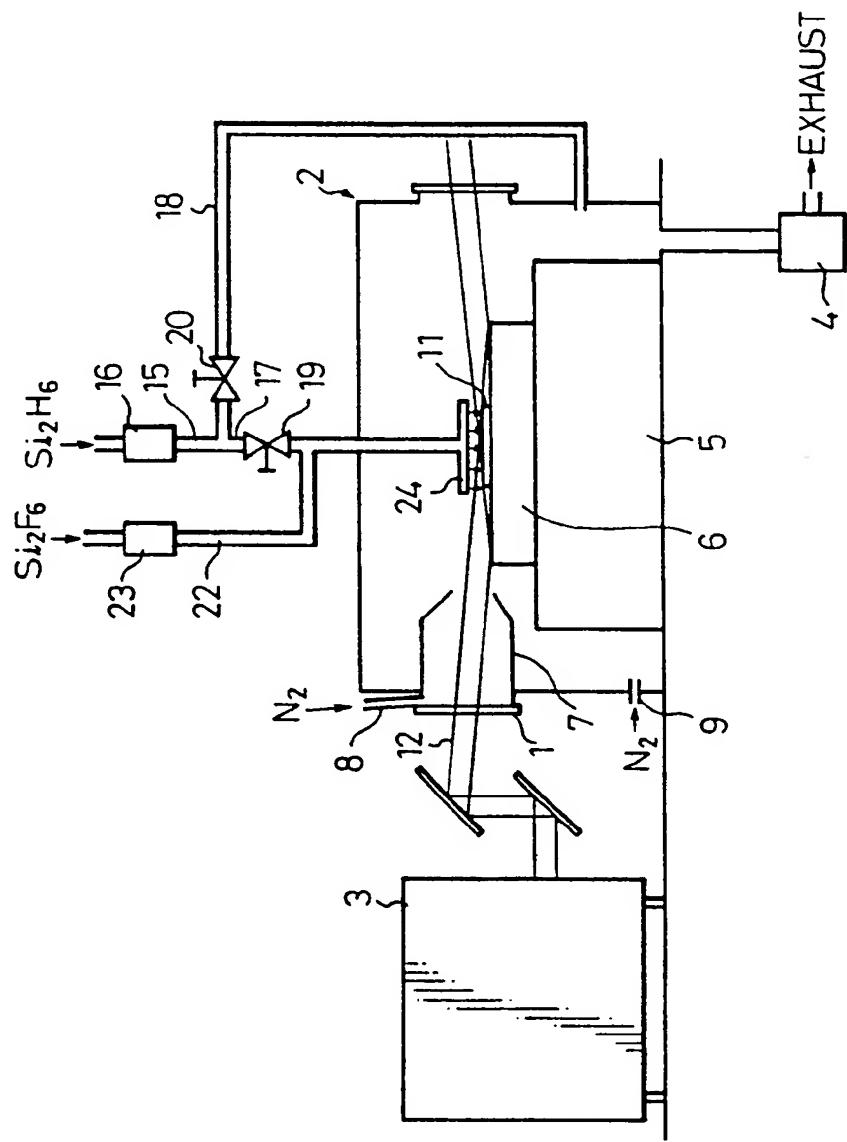


Fig. 2

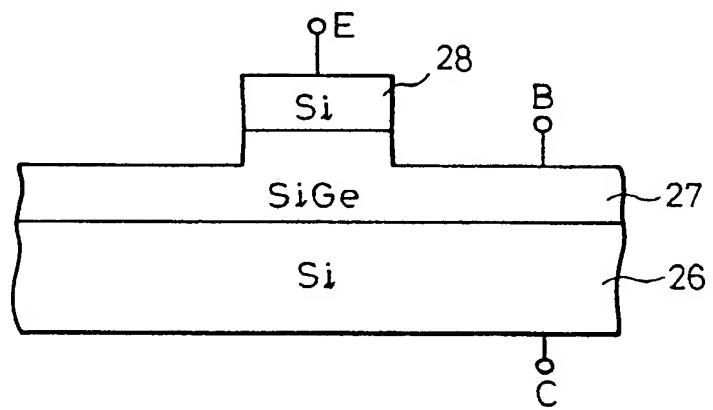
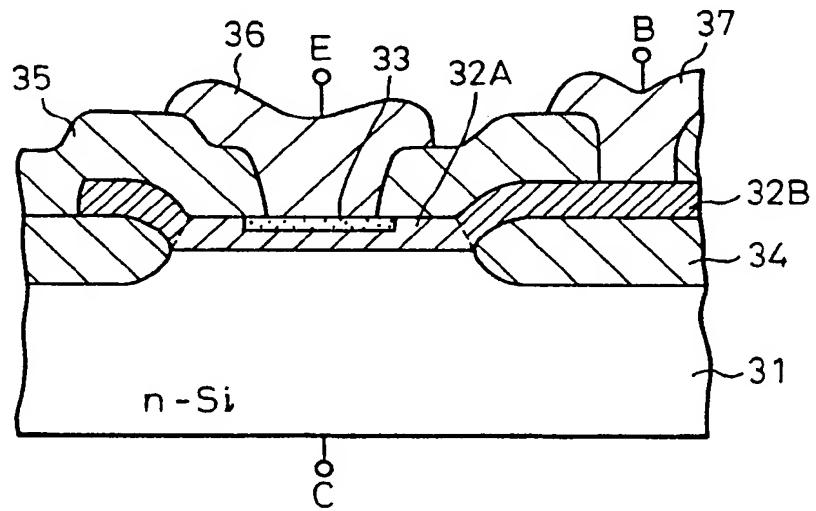
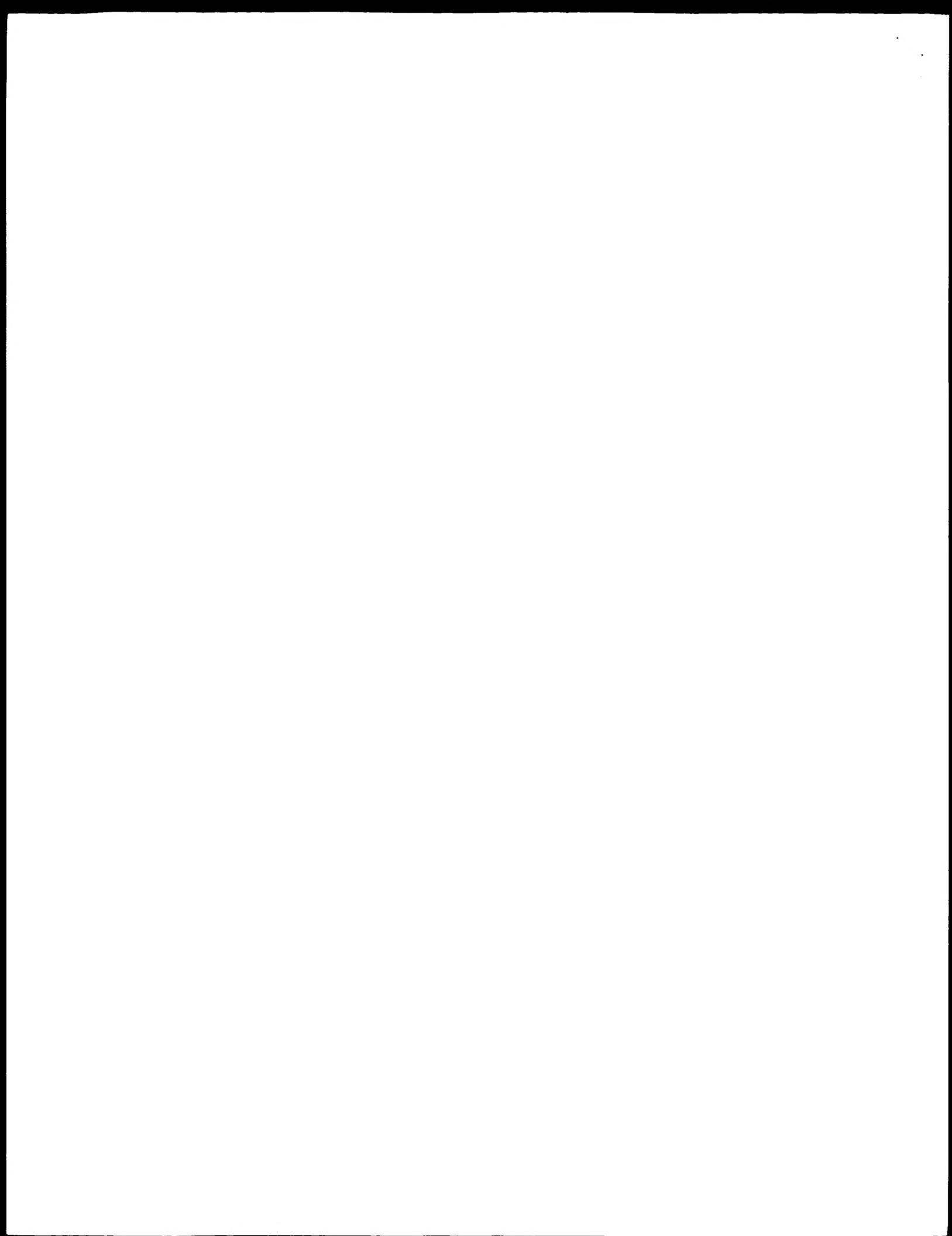


Fig. 3







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Prior to the epitaxial growth of the semiconductor layer, the substrate is cleaned by allowing the fluoride gas to flow over the substrate having a temperature of from a room temperature to 500 °C, and by irradiating an ultraviolet light to the substrate to remove a natural oxide layer from the substrate surface.

EP 0 368 651 A3



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DOCUMENTS CONSIDERED TO BE RELEVANT			
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int. Cl. 5)
X	PATENT ABSTRACTS OF JAPAN, vol. 11, no. 273 (C-445)[2720], 4th September 1987; & JP-A-62 78 191 (MITSUI TOATSU CHEM. INC.) 10-04-1987 * The whole abstract * ---	1-6	C 30 B 25/10 C 30 B 29/06
X	PATENT ABSTRACTS OF JAPAN, vol. 11, no. 273 (C-445)[2720], 4th September 1987; & JP-A-62 78 192 (MITSUI TOATSU CHEM. INC.) 10-04-1987 * The whole document * ---	1-6	
X	PATENT ABSTRACTS OF JAPAN, vol. 11, no. 273 (C-445)[2720], 4th September 1987; & JP-A-62 78 193 (MITSUI TOATSU CHEM. INC.) 10-04-1987 * The whole abstract * ---	1-6	
A	PATENT ABSTRACTS OF JAPAN, vol. 112, no. 237 (C-509)[3084], 6th July 1988; & JP-A-63 0398 (TOSHIBA CCRP.) 09-02-1988 * The whole abstract * ---	1-8	
X, D	JAPANESE JOURNAL OF APPLIED PHYSICS; SUPPLEMENTS (18TH INT. CONF. ON SOLID STATE DEVICES AND MATERIALS, TOKYO 1986), 20th - 22nd August 1986, pages 217-220, Tokyo, JP; A. YAMADA et al.: "Photochemical vapor deposition of single-crystal silicon at a very low temperature of 200 degrees C" * Page 217 * ---	1-6, 9	C 30 B C 23 C
A	DE-A-3 237 338 (FRAUNHOFER-GESELLSCHAFT ZUR FÖRDERUNG DER ANGEWANDTEN FORSCHUNG e.V.) ---	-/-	
<p>The present search report has been drawn up for all claims</p>			
Place of search	Date of completion of the search	Examiner	
THE HAGUE	31-08-1990	COOK S.D.	
CATEGORY OF CITED DOCUMENTS		<p>T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons &amp; : member of the same patent family, corresponding document</p>	
<p>X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document</p>			



DOCUMENTS CONSIDERED TO BE RELEVANT			CLASSIFICATION OF THE APPLICATION (Int. Cl.5)
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	
X	APPLIED PHYSICS LETTERS, vol. 49, no. 2, July 1986, pages 79-81, American Institute of Physics, New York, US; S. NISHIDA et al.: "Epitaxial growth of silicon by photochemical vapor deposition at a very low temperature of 200 degrees C" * Page 79 *	1-6, 9	
A	IBM TECHNICAL DISLCOSURE BULLETIN, vol. 6, no. 9, February 1964, pages 45-46, New York, US; G. CHEROFF et al.: "Reversible and nonreversible light stimulated deposition of Ge and Si"	---	
A	US-A-4 728 528 (ISHIHARA et al.)	-----	
			TECHNICAL FIELDS SEARCHED (Int. Cl.5)
<p>The present search report has been drawn up for all claims</p>			
Place of search	Date of completion of the search	Examiner	
THE HAGUE	31-08-1990	COOK S.D.	
CATEGORY OF CITED DOCUMENTS		T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons ..... & : member of the same patent family, corresponding document	
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